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High Anionic Conductive Form of $\text{Pb}_x\text{Sn}_{2-x}\text{F}_4$

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Abstract

High anionic conductivity of $\sim 3.5 \times 10^{-3} \text{ Scm}^{-1}$ at room temperature is achieved for $\text{Pb}_x\text{Sn}_{2-x}\text{F}_4$ ($x = 1.21$) obtained by annealing a mechanically-milled $\text{PbF}_2/\text{SnF}_2$ mixture at 400 °C. The observed synchrotron X-ray diffraction patterns indicate formation of a new tetragonal phase at $x = 1.1 \sim 1.3$. The Rietveld analysis of the neutron diffraction patterns leads to a unique structure consisting of two alternating layers; a double Pb layer and a triple layer, each flanked by a single Sn layer. As the Rietveld analysis does not fully converge, the authors further apply high-resolution solid-state NMR (^{19}F , ^{119}Sn , and ^{207}Pb) to confirm the structure. Further the ^{19}F - ^{207}Pb cross-polarization experiment shows that most of F^- ions, except for those lie between the double Pb layers, contribute to its high ionic conductivity. The high conductivity is also attributed to structural flexibility of the triple Pb layers, indicated by temperature dependent ^{207}Pb NMR spectra.

Keywords: solid-state NMR, solid electrolyte conductive for F^- , synchrotron XRD, neutron diffraction

1. Introduction

Solid electrolytes conductive for F^- are attracting strong attention as one of promising electrolytes for new batteries using fluoride compounds as positive and negative active materials, which are expected to give high energy density.¹⁻¹² Among different kinds of solid electrolyte conductive for F^- , $PbSnF_4$ has long been known to show very high conductivity.¹³ Causes of high ionic conductivity of the material are not completely understood yet and structural studies have been done to elucidate the relationship between its structure and the conduction properties.^{14,15} Different methods have been applied to prepare $PbSnF_4$, e.g. mechanical milling, precipitation reaction, and solid-state reaction of a mixture of PbF_2 or $Pb(NO_3)_2$ and SnF_2 followed by annealing in inert gas.^{16,17} It has been shown that conductivity depends on its crystal morphology. Unfortunately, the definition of the name of these forms is not common among references, then the authors adopted the one defined in refs. [18] in this work, where the main three forms, monoclinic, tetragonal, and cubic forms, are referred as to the α -form, the β -form, and the γ -form, respectively. Its polymorphism scheme is roughly written as follows; the α -form obtained at lower annealing temperature below ca. 80 °C is converted to the β -form at above 80 °C and further to the other tetragonal form (β') at ~ 350 °C and finally to the γ -form at ~ 380 °C. Recently, it was shown that the γ -form can also be obtained by mechanical milling without thermal annealing.^{19,20} Except for the cubic γ -form, in which Pb and Sn occupy the cubic lattice positions, the alignment of the two cations, Pb and Sn, are common for α and β , that is, double Pb layers are flanked by a single Sn layer, which may be schematically written as $[-Sn-Pb-Pb-Sn-]$, along the c axis.

So far, various studies have been done to improve its conductivity. For example, Uno et al. examined β - PbSnF_4 obtained by mechanical milling of PbF_2 and SnF_2 powders with a certain ratio of the materials.²¹ The conductivity depends on the materials ratio and is largest ($\sim 1.6 \times 10^{-3} \text{ Scm}^{-1}$ at room temperature) at $\text{PbF}_2:\text{SnF}_2=50:50$. Interestingly, the conductivity can be improved by ca. 1 order of magnitude by annealing the pellet in the platinum tube under nitrogen gas at 400 °C for 6 hrs. They attributed the increase to the reduction of grain-boundary impedance due to melting of grains.

This work exploits a Pb-rich region aiming at the higher conductivity; a new crystalline form may appear by annealing at a higher temperature allowed by higher melting point of Pb-rich compounds. Indeed, in this work, it is shown that the annealing temperature of 400 °C leads the high conductivity of $\text{Pb}_x\text{Sn}_{2-x}\text{F}_4 \sim 3.5 \times 10^{-3} \text{ Scm}^{-1}$ at $x \sim 1.21$, which is ca. 1.6 times higher than that of the β -form.

To elucidate the structural origin of the high conductivity, various kinds of technology have been employed in this work; (1) synchrotron X-ray powder diffraction (SXRD), (2) neutron diffraction (ND), and (3) high-resolution solid-state NMR for ^{207}Pb , ^{119}Sn , and ^{19}F . Combined analysis using these technologies leads to detailed understanding of the structure of the new tetragonal $\text{Pb}_x\text{Sn}_{2-x}\text{F}_4$ form at $x = 1.1\sim 1.3$. The material is referred as the β^+ -form from now on. The β^+ -form is composed of an alternating two-layers structures; one is the double Pb layer $[-\text{Sn-Pb-Pb-Sn-}]$, whose structure is similar to that in the β -form. The other is a triple Pb layer, which is written as $[-\text{Sn-Pb-Pb-Pb-Sn-}]$. Experimental results using NMR show that the F^- ions between the two Pb layers in the double Pb layer are less mobile but that F^- ions among the Pb layers in the triple Pb layer are more mobile. The

latter F^- ions causes the higher conductivity of the β^+ -form. Further, the temperature-dependent NMR chemical shift of ^{207}Pb suggests high flexibility of the triple Pb layer.

2. Experimental Section

2.1 Sample Preparation

Several $\text{Pb}_x\text{Sn}_{2-x}\text{F}_4$ samples ($x = 1.52\sim 0.96$) were prepared from mixture of PbF_2 (Kojundo Chemical Laboratory Co., Ltd., 99.9%) and SnF_2 (Kojundo Chemical Laboratory Co., Ltd., 99.9%) powders as the starting materials. A mixture of these materials in different ratio was mechanically milled at room temperature by a planetary ball mill using a zirconia pot with 10 zirconia balls. The rotation speed was 600 rpm and the milling time was 6 h. The prepared amorphous powders were annealed at 400 °C for 1 h in an alumina crucible. The high annealing temperature 400 °C, which is close to the melting point (ca. 390 °C) of $\text{Pb}_x\text{Sn}_{2-x}\text{F}_4$ with $x \sim 1$, was selected not for melting the grain boundary in order to enhance ionic conductivity of the pellet but for leading to the formation of a new phase of high conductivity for F^- in mechanically milled powder. In fact, all samples obtained after annealing at 400 °C were smooth powder, and reduction of grain-boundary impedance due to melting is thus less likely for all samples examined. The sample prepared from a 1:1 mixture was annealed at 200 °C to prepare the PbSnF_4 sample in the β -form; β - PbSnF_4 sample, hereafter. All the processes were performed in a dry Ar atmosphere.

The mole ratios of Pb and Sn ($\text{Pb}:\text{Sn} = x:2-x$) in the annealed samples were determined by using inductively coupled plasma spectrometer (ICP, ICP-8000E, Shimadzu). The mole

ratios in the β -PbSnF₄ sample were found to be 0.96:1.04, and the ratios for the samples annealed at 400 °C are designated by the x value.

Electrical conductivities were measured for the cold-pressed pelletized samples with 10 mm in diameter and about 1 mm in thickness (200 mg). Gold electrodes were formed on both sides of samples. AC impedance measurements were carried out in a dry Ar atmosphere using an impedance analyzer (SI1260, Solartron) in the frequency range of 0.1 Hz to 1 MHz. The temperature range of the measurements was from -35 to 140°C.

2.2 Synchrotron X-ray powder diffraction (SXRD)

Synchrotron X-ray powder diffraction (SXRD) measurements were performed at an undulator beamline, BL28XU in SPring-8, Japan in the transmission geometry. The powder samples were sealed in a borosilicate glass capillary (wall thickness = 0.01 mm and outer diameter = 0.3 mm) and rotated by a spinner during the measurement. The incident X-ray monochromated by a Si (220) channel-cut monochromator with a wavelength of $\lambda = 0.050043$ nm (calculated by using CeO₂ as a standard) was used with a beam size of 0.5 mm by 0.5 mm shaped by a quadrant slit. The intensity of the incident X-ray was monitored by an ionization chamber with N₂ and Ar gas flow located upstream of the sample, used for the normalization of the XRD data. The X-ray transmitted from the sample was blocked by a lead direct stopper located downstream of the sample to suppress the air scattering. The diffraction patterns were measured using a two-dimensional pixel array detector, Pilatus 100K (Dectris), mounted on the 2θ arm of the goniometer with the camera length of ~1000 mm. The angle conversions from the diffraction images to XRD profiles were carried out

by integration along a circular arc with the histogram method²² in combination with a 3 x 3 super-sampling method.

2.3 Neutron diffraction (ND)

Neutron diffraction data were collected on the time-of-flight neutron diffractometer, SPICA at MLF (Materials and Life science experimental Facility) in J-PARC (Japan Proton Accelerator Research Complex).²³ The sample was installed in a cylindrical 6 mm diameter vanadium can. The range of wavelength of the neutron beam, which can be used for analysis, was from 0.05 to 0.98 nm. The diffraction pattern was collected using all detectors in the range of $10^\circ < 2\theta < 175^\circ$. The 90° bank ($70^\circ < 2\theta < 110^\circ$) data, which covered the d space range from 0.03 to 0.58 nm, were used for the analysis in this work. The structural parameters were refined with the program, Z-Rietveld.²³

2.4 NMR measurement

The NMR measurements were made using a JEOL ECA600 NMR spectrometer with a triple-tuned MAS probe (Agilent Technologies Inc.) for a 1.6 mm rotor at 14 T. The resonance frequencies for ^{19}F , ^{119}Sn , and ^{207}Pb were 564 MHz, 224 MHz, and 126 MHz, respectively. The ^{19}F chemical shifts were calibrated in ppm relative to CCl_3F adopting the ^{19}F chemical shift for LiF (-203 ppm²⁴) as an external reference. The ^{119}Sn chemical shifts were calibrated in ppm relative to $\text{Sn}(\text{CH}_3)_4$ by adopting the ^{119}Sn chemical shift for SnO_2 (-603 ppm²⁵) as an external reference. ^{207}Pb chemical shifts were calibrated in ppm relative

to saturated aqueous solution of $\text{Pb}(\text{NO}_3)_2$. The temperature-calibration experiment was done using ^{207}Pb NMR of $\text{Pb}(\text{NO}_3)_2$.²⁶

The MAS NMR spectra were observed by using a single-pulse method or a cross-polarization (CP) method. Except for ^{19}F experiments, the Hahn echo was used with the echo-delay time being $1/\nu_R - pw/2$, where ν_R is the MAS spinning frequency and pw is the π pulse length. The CP experiment between ^{19}F and ^{207}Pb spins was made by the conventional constant-amplitude CP pulse sequence. The rf amplitudes used in CP was ca. 40 kHz for ^{19}F and ca. 33 kHz for ^{207}Pb , and the contact time of 0.8 ms was used. The ^{207}Pb spin-lattice relaxation times in the rotating frame ($T_{1\rho}$) under the same rf amplitude were measured under MAS by using the conventional spin-lock method with the recovery data being fitted to a single exponential curve. To examine the distance information among the Pb sites, we applied the two-dimensional (2D) Radio Frequency Driven Recoupling (2D-RFDR).²⁷ It was performed with an experimentally optimized π pulse length of 6.8 μs using XY16 phase cycling and the mixing time of 9.6 ms. The 2D RFDR spectrum was observed using 50 t_1 increments and 128 t_2 complex points. The experimental data sets were zero-filled in both the t_1 and t_2 dimensions to form a 1024×1024 data matrix before 2D Fourier Transformation with 200 Hz exponential broadening. The MAS frequency was 20 kHz for RFDR and 25 kHz for other experiments

3. Results and Discussion

For the $\text{Pb}_x\text{Sn}_{2-x}\text{F}_4$ ($x = 0.96\sim 1.52$) samples annealed at 400 °C, its observed conductivity at room temperature (Figure 1) depends on the mole ratios (x) of Pb and Sn. Note that the conductivity for the $\beta\text{-PbSnF}_4$ sample ($\sim 2.2 \times 10^{-3} \text{ Scm}^{-1}$) is comparable to that reported previously for $\beta\text{-PbSnF}_4$ prepared without annealing ($\sim 1.6 \times 10^{-3} \text{ Scm}^{-1}$)²¹ and $\beta\text{-PbSnF}_4$ prepared with annealing at 200 °C ($\sim 1.6 \times 10^{-3} \text{ Scm}^{-1}$).¹⁹ The conductivity values of the samples with $x = 1.15\sim 1.25$ are higher than that of the $\beta\text{-PbSnF}_4$ sample. The sample prepared by cold pressing of the $\text{Pb}_x\text{Sn}_{2-x}\text{F}_4$ powder showed the ionic conductivity as high as $\sim 3.5 \times 10^{-3} \text{ Scm}^{-1}$ at $x = 1.21$. The value was ca. 1.6 times higher than that of the β -form at $x = 0.96$.

Figure 2 shows the SXRD patterns of the $\text{Pb}_x\text{Sn}_{2-x}\text{F}_4$ ($x = 0.96\sim 1.52$) samples annealed at 400 °C and that of the $\beta\text{-PbSnF}_4$ sample ($x = 0.96$). The observed $\beta\text{-PbSnF}_4$ pattern is consistent with those reported previously.^{16,28} The characteristic SXRD pattern at $2\theta = 13\sim 17^\circ$ of the $x = 1.52$ sample annealed at 400 °C indicates that this sample takes the γ -form,²⁸ and the $x = 0.96$ sample is the α -form.¹⁶ It is worthy to note that, similar to the α -form obtained at lower annealing temperature below ca. 80 °C, the present α -form showed phase transition to the β -form at above 80 °C. This phase transition behavior also shows that the two α -forms are same. However, it should be mentioned that reproducibility of the formation of the pure α -form is not good as compared to those of the other forms. In fact, many samples starting from the Pb:Sn = 1:1 mixture bear the SXRD patterns consisted of the α - and β/β^+ -forms.

The SXRD patterns of the $x = 1.15\sim 1.25$ samples, whose conductivities are higher than that of the $\beta\text{-PbSnF}_4$ sample (Figure 1), resemble the SXRD pattern of the $\beta\text{-PbSnF}_4$

sample. Detailed analysis of the patterns, however, shows that there is an apparent shift of the peak at around $2\sim 3^\circ$ designated by a round-square box in Figure 2; the peak appeared at ca. 2.5° for the α - and β -forms moves toward the lower angle (ca. 2.2°) for the $x = 1.15\sim 1.25$ samples. This indicates elongation of the c axis in these samples, in other words, the structure of these samples (the β^+ -form) is new.

In summing the observed relationship between the crystalline form and its conductivity at room temperature, the highest conductivity $\sim 3.5 \times 10^{-3} \text{ Scm}^{-1}$ should be ascribed to for β^+ at $x = 1.21$, while the β -form bears slightly lower conductivity of $\sim 2.2 \times 10^{-3} \text{ Scm}^{-1}$. The conductivity for the other two forms (α and γ) is much lower; $\sim 0.93 \times 10^{-3} \text{ Scm}^{-1}$ for the α -form and $\sim 0.83 \times 10^{-3} \text{ Scm}^{-1}$ for the γ -form. It is of interest to examine the structural feature of the β^+ -form in order to understand its high conductivity.

The neutron diffraction (ND) experiment was applied to examine its structure. The structural parameters were refined with space group $P4/nmm$ (129) by the Rietveld method using the program, Z-Rietveld. Among the three β^+ samples ($x = 1.15, 1.21, 1.25$), the best fitting was obtained for the $x = 1.15$ sample. However, the fitting was still not good enough to lead a fully converged structure. The Rietveld refinement pattern of the β^+ -PbSnF₄ sample ($x = 1.15$) at room temperature is shown in Supporting Information as Figure S1, and the packing of β^+ -PbSnF₄ in the unit cell is shown in Figure 3. It should be pointed out here that the Rietveld analysis of the SXRD data also gave a structure (not converged) almost identical to that shown in Figure 3. As the R factors for fitting of the ND data are rather small ($R_{\text{wp}} = 0.08388$, $R_e = 0.04018$, $R_F = 0.21109$), we proceeded with the deduced structure for further examination.

It is suggested that the β^+ -form consists of two alternating layers; one layer resembles to the double layer of the β -form (the region designated by a two-way arrow in Figure 3), while the other layer has a triple Pb layer flanked by a single Sn layer. The calculated chemical formula is $\text{Pb}_5\text{Sn}_4\text{F}_{18}$ ($Z = 1$). For direct comparison with the β -form, we may write this as $\text{Pb}_{1.11}\text{Sn}_{0.89}\text{F}_4$ ($Z = 4.5$). In the following, we refer simply to it as β^+ - PbSnF_4 . Note that the calculated x value for the β^+ -form is $x = 1.11$, which is close to the x value of the sample ($x = 1.15$) used for the ND analysis. Interestingly, better conductivity is achieved for a slightly larger value of $x = 1.21$, indicating enhancement of conductivity by structural disorder brought about by additional Pb. One possible assignment of the excess Pb may be an extra triple Pb layer; the layer structure can be written as $-[3]-[2]-[3]-[3]^*- [2]-[3]-$, where $[n]$ exhibits the n -Pb layer with the flanking Sn layers, and $[3]^*$ indicates the extra triple Pb layer. Such disorder in a long-range structure would disturb the Rietveld analysis of the ND experiment. Further, the Rietveld fitting becomes difficult because of large anisotropic thermal parameters of F atoms reflecting its facile motion and also of partial site occupancies, which enable efficient defect diffusion²⁹ (ion conductivity). In this work, to confirm the local structure of β^+ - PbSnF_4 suggested by the analysis of the SXRD and ND results, we apply high-resolution solid-state NMR, which is a powerful mean to examine the local structure without being disturbed by the long-range disorder. All constitute elements of PbSnF_4 are NMR active (^{19}F , ^{119}Sn , and ^{207}Pb), and among them, ^{207}Pb -NMR would be of interesting as three different Pb sites, namely, two sites in the triple Pb layer and one in the double Pb layer are suggested. In the following NMR measurement, we examine the $x = 1.21$ sample as the β^+ - PbSnF_4 sample and compare

with the results of β -PbSnF₄.²⁹

Figure 4 compares ^{19}F , ^{119}Sn , and ^{207}Pb MAS NMR spectra of the two samples (β and β^+). All spectra were taken by using the single-pulse sequence without the Hahn echo for ^{19}F , and with the Hahn echo for ^{119}Sn and ^{207}Pb . The ^{207}Pb spectrum of the β^+ sample taken by using ^{19}F to ^{207}Pb CP is also given, which will be discussed afterward.

As there are several crystallographically different F sites in the β - and β^+ -forms, one may naively expect to find several signals in the ^{19}F MAS spectra. However, the two samples exhibit only one relatively sharp ^{19}F signal, which is ascribable to motional averaging of the chemical-shift differences among the different F sites. The narrower linewidth of the ^{19}F signal of the β^+ sample may indicate faster ^{19}F motion in the β^+ sample. The origin of the apparent chemical-shift difference between the two ^{19}F spectra should be attributable to differences of the local structures in the two forms.

Similar difference of the isotropic shifts is also observed for the ^{119}Sn spectra in Figure 4, in which the spinning sidebands marked by asterisks appear. It is notable that the MAS spinning side-band patterns are quite similar for these two samples, which indicates similar local environment for the Sn nuclei in the two forms. The characteristic MAS pattern was attributed to the local coordination structure of Sn in the β -form; four $4f$ F atoms and one $2c$ F atom are placed one-sidedly thus leading the large chemical-shielding anisotropy. Detailed analysis of the ^{119}Sn lineshape of the β -form including effects of F^- motion is found in Ref. [29]. To conclude, the almost same sideband patterns for the β and β^+ -forms suggest an almost identical local coordination structure of Sn for them, which is consisted with the structure obtained by the ND experiments.

In contrast to the ^{19}F and ^{119}Sn results, there is salient difference in the ^{207}Pb spectra; there are three ^{207}Pb signals in the spectrum of the β^+ sample, while only one signal is obtained for the β sample. For ease of discussion, the three signals are numbered as "a" to "c" from the higher-frequency side (Figure 4) and the corresponding ^{207}Pb nuclei as Pb-a to Pb-c. The area-intensity ratio for the three signals obtained by lineshape decomposition (Figure 5) is 40.2:19.7:40.1~2:1:2. It appears that the ^{207}Pb signal-c at ca. -75 ppm corresponds to the ^{207}Pb signal in the β -form, and we assign Pb-c to Pb (Pb5) in the double Pb layer in the β^+ -form (Figure 3).

The remaining two Pb signals are assigned to the two Pb nuclei in the triple Pb layer, and, with the ratio obtained from the Pb-signal intensities of Pb-a:Pb-b = 2:1, we assign Pb-a to Pb2 and Pb4 sitting at the outside of the triple Pb layer and Pb-b to Pb3 at the inside of the layer. To confirm Pb-a and Pb-b are in the same triple layer, we applied ^{207}Pb - ^{207}Pb dipolar-correlation experiment by using the RFDR method.

Figure 6 shows the ^{207}Pb - ^{207}Pb 2D-RFDR spectrum of the β^+ -PbSnF₄ sample taken with the mixing time of 9.6 ms and the MAS spinning speed of 20 kHz. Even though the natural abundance of ^{207}Pb is low (ca. 21%), we found cross peaks between Pb-a and Pb-b. The observation of the cross peaks indicates that Pb-a and Pb-b are in close proximity, which confirms the above assignment, namely, Pb-a and Pb-b belong to the triple Pb layer. To conclude, Pb-a is assigned to the outer Pb in the triple Pb layer (Pb2 and Pb4) and Pb-b to the inner one (Pb-3). The Pb-a:Pb-b:Pb-c = 2:1:2 ratio obtained from the ^{207}Pb spectrum (Figure 5) further requires the ratio of the triple and the double layers is close to 1:1, which is indeed consistent with the alternating structure in Figure 3. It is worthy to point out here

that the three ^{207}Pb signals were also observed for the other two β^+ samples ($x = 1.15$ and 1.25).

The Pb-c signal disappears in the 2D-RFDR spectrum. This is explained as follows. The broad linewidth of the ^{207}Pb -c signal is ascribable to inhomogeneous one due to distribution of local environment of Pb-c. The Pb-Pb magnetization transfer occurs among the different Pb-c sites during the mixing time in the RFDR experiment, which leads to scattering of the Pb intensity among the broad signal, i.e., the observed reduction of the signal.

Having established the structure of the β^+ -form, let us now turn to the relationship between its unique structure and high conductivity. It is apparent that facile transportation of F^- ions is supported by its unique Sn-Pb layer structure with the "gap" between the Pb layer and the Sn layer that offers a conduction path. The fast F^- motion should reduce the ^{19}F - ^{207}Pb dipolar interactions and it was in fact observed that ^{19}F dipolar decoupling brings no apparent line narrowing for the ^{207}Pb MAS spectra (not shown). As the strength of the ^{19}F - ^{207}Pb dipolar interactions at the three different Pb sites is a good measure of the mobility of the F^- ions in the conduction path, that is, through the double Pb layer and through the triple Pb layers, the ^{19}F to ^{207}Pb CP experiments were pursued; the CP transfer rate is sensitive to the strength of the ^{19}F - ^{207}Pb dipolar interaction. It is remarkable that only the Pb-c signal does appear (the red spectrum in Figure 4f). This means that the ^{19}F - ^{207}Pb dipolar interaction among ^{19}F and ^{207}Pb at Pb-a and Pb-b in the triple layer is weak and/or the ^{207}Pb spin-lattice relaxation time in the rotating frame ($T_{1\rho}$) is short. The latter is negated because the observed $T_{1\rho}$ values for Pb-a and Pb-c (~ 10 ms) are much longer than

that for Pb-c (~ 3 ms) and the contact time for the CP experiment (0.8 ms). The result of the CP experiment thus leads us to conclude that the F^- ions around Pb-a and Pb-b are mobile, and the ones around Pb-c are less mobile. Additionally, the ^{119}Sn - ^{19}F CP experiment for the β^+ -form did not give any ^{119}Sn signals (not shown). These data indicate that all F ions are mobile except for F8(2b) in the β^+ -form, that is, the ratio of the mobile F^- ions in β^+ -form is much larger than that in the β -form. This high concentration of mobile F^- leads to the higher conductivity of the β^+ -form.

To examine further the origins of the higher conductivity of these tetragonal forms, temperature dependence of the ^{207}Pb MAS spectrum was examined. Temperature dependence of the ^{19}F and ^{119}Sn spectra is in due course and detailed ion dynamics will be discussed. Figure 7 shows the temperature dependence of the ^{207}Pb MAS spectra of the β^+ -form and the β -form. Indeed, all signals show lower-frequency shift at higher temperature. Similar to the spectral decomposition in Figure 5, the β^+ -form spectra were fitted to a sum of the three signals. The chemical shifts at each temperature for the three ^{207}Pb signals of the β^+ -form thus obtained are plotted with that of the β -form in Figure S2 of the Supporting Information. It is shown that, in the β^+ -form, the Pb-a signal in the triple Pb layer shifts more than the other Pb signals. This indicates that the triple Pb layer is more flexible than the double Pb layer. The shift of the Pb signal in the β -form is comparable to those of Pb-b and Pb-c in the β^+ -form. These results indicate that the flexibility of the Pb layers in both of the β -form and the β^+ -form alleviates steric hindrance for conduction of F^- ions in the layer and leads to higher conductivity in β^+ -form.

4. Summary

A new compound of $\text{Pb}_x\text{Sn}_{2-x}\text{F}_4$ ($x \sim 1.21$) with a layered crystalline structure was prepared using the high annealing temperature of 400 °C. The SXRD study suggested that the compound is in the tetragonal structure and has a longer c axis as compared to that of the reported tetragonal form, which has been described as the β -form.^{16,28} The new form named here as the β^+ -form showed the anionic conductivity at room temperature as high as $\sim 3.5 \times 10^{-3} \text{ Scm}^{-1}$ for the $x = 1.21$ sample, that is appreciably higher than the value obtained for the other form of PbSnF_4 ; $\sim 0.93 \times 10^{-3} \text{ Scm}^{-1}$ for the monoclinic form, $\sim 2.2 \times 10^{-3} \text{ Scm}^{-1}$ of the tetragonal β -form, and $\sim 0.83 \times 10^{-3} \text{ Scm}^{-1}$ for the cubic form. Note that the conductivity of $2.5 \times 10^{-3} \text{ Scm}^{-1}$ was reported recently for the Pb-rich ($x \sim 1.1$) cubic form obtained by ball milling without following thermal annealing.²⁰

The Rietveld analysis of the neutron diffraction (ND) profile and the synchrotron X-ray diffraction (SXRD) profile did not bring out a fully converged structure. High-resolution solid-state NMR studies of ^{19}F , ^{119}Sn , and ^{207}Pb of the materials gave a lot of information of the structure of the material; (1) the local environment of Sn in the β^+ -form is almost identical to that in the β -form, (2) there are three Pb sites in the β^+ -form and one of them is attributable to Pb in a site similar to the Pb site in the double Pb layer in the β -form, and (3) the ratio of the three Pb sites obtained from the Pb-signal intensities is 2:1:2. These are in fact consistent with the structure derived from the Rietveld analysis of the ND data, that is, an alternating layered structure with the double and the triple Pb layers.

As for the higher conductivity for β^+ , several experimental clues were shown; (1) The

^{19}F - ^{207}Pb CP magnetization transfer did not occur for the two ^{207}Pb sites in the triple Pb layer. (2) The highest conductivity was achieved for $\beta^+\text{-Pb}_x\text{Sn}_{2-x}\text{F}_4$ ($x = 1.21$), whose x value is slightly larger than the ideal one for the β^+ -form ($x = 1.11$). (2) The chemical shifts of the ^{207}Pb signals of the β -form and the β^+ -form are temperature dependent. The first point is that the better conductivity of the β^+ -form is ascribed to the higher mobility of the F^- ions in the triple Pb layer. The second point is that the structural disorder brought about by the excess Pb, that is considered to increase the amount of the triple layer. This extra triple layer leads to higher conductivity for $\beta^+\text{-Pb}_x\text{Sn}_{2-x}\text{F}_4$ at $x = 1.21$. The last point is that the structural flexibility of the conduction pathways allows facile ion conduction in these tetragonal PbSnF_4 forms in addition to the defect-driven conductivity shown for the β -form.²⁹ The present results may invoke a similar enhancement of conductivity based on flexibility of the conduction pathway postulated for $\text{LiP}_3\text{S}_{11}$ metastable crystal.³¹

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Supporting Information Available: Observed and calculated neutron diffraction profile for β^+ -PbSnF₄ ($x = 1.14$) was given in Figure S1. Temperature dependence of the ^{207}Pb chemical shifts in β^+ - and β -PbSnF₄ obtained by spectral decomposition was given in Figure S2. This material is available free of charge via the Internet at <http://pubs.acs.org>.

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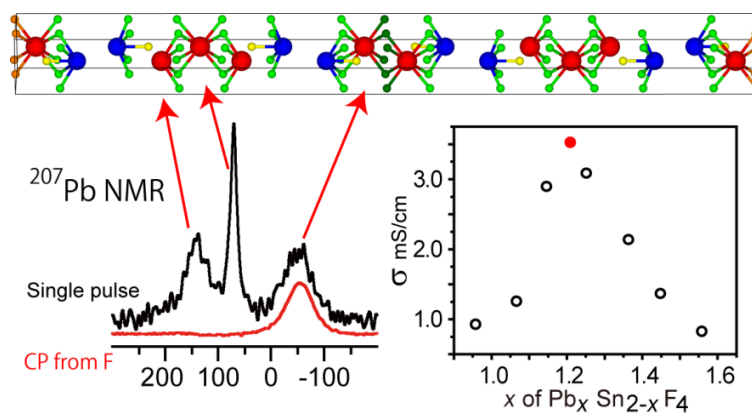
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TOC Abstract Graphics

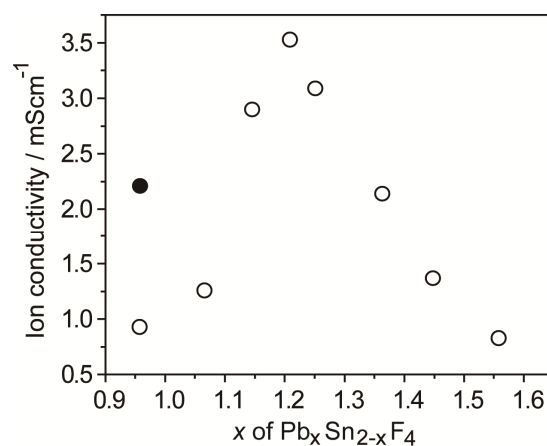


Figure 1. Dependence of conductivity on the Pb:Sn molar ratio in the $\text{Pb}_x\text{Sn}_{2-x}\text{F}_4$ pressed-powder samples ($x = 0.96 - 1.56$) annealed at 400 °C (open circle) and 200 °C (filled circle).

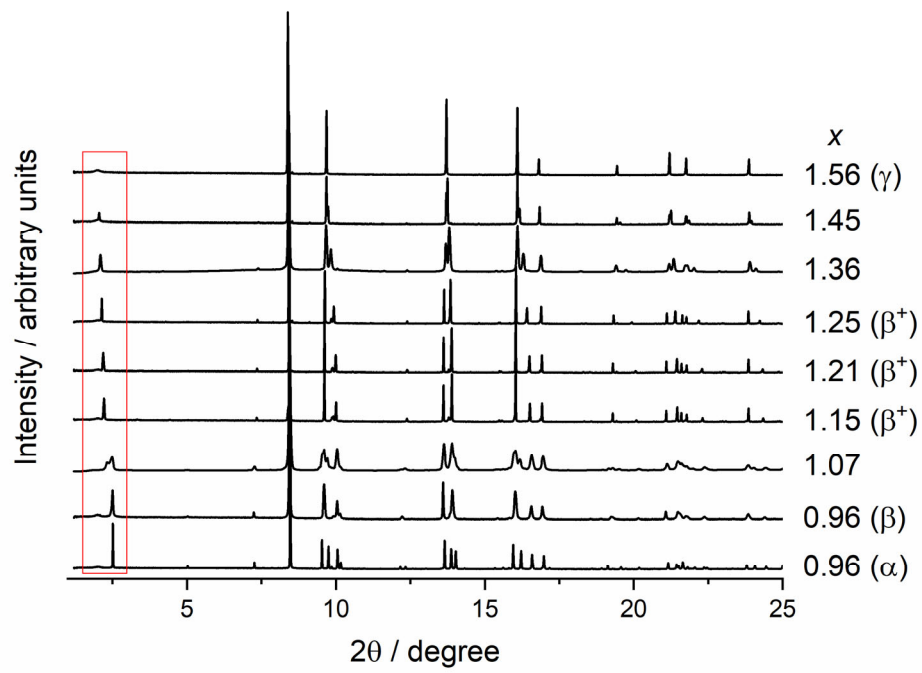


Figure 2. SXRD patterns of the $\text{Pb}_x\text{Sn}_{2-x}\text{F}_4$ ($x = 0.96 - 1.56$) samples. The patterns without labeling are of the transition/mixed states.

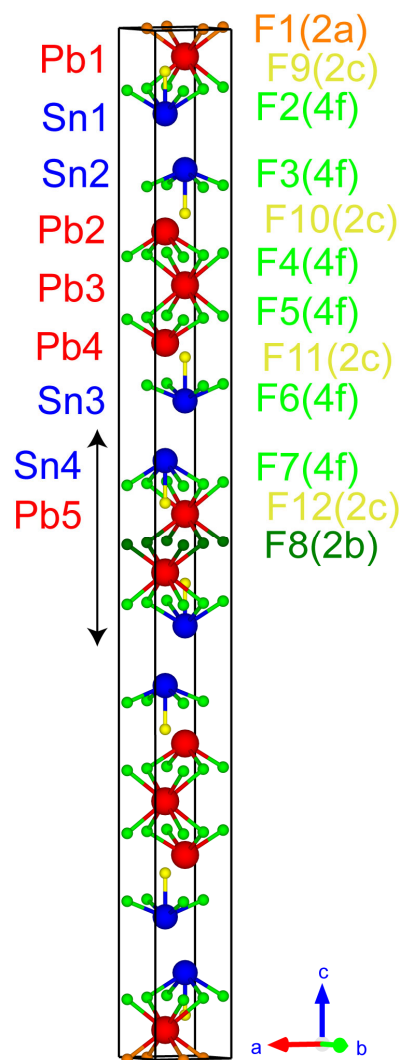


Figure 3. Packing of β^+ -PbSnF₄ in the unit cell. *a*, *b*, and *c* denotes three crystallographic axes. Blue and red balls are Sn and, Pb respectively. The four F sites (2a, 2b, 2c, and 4f) are designated by orange, dark-green, yellow, and green, respectively.

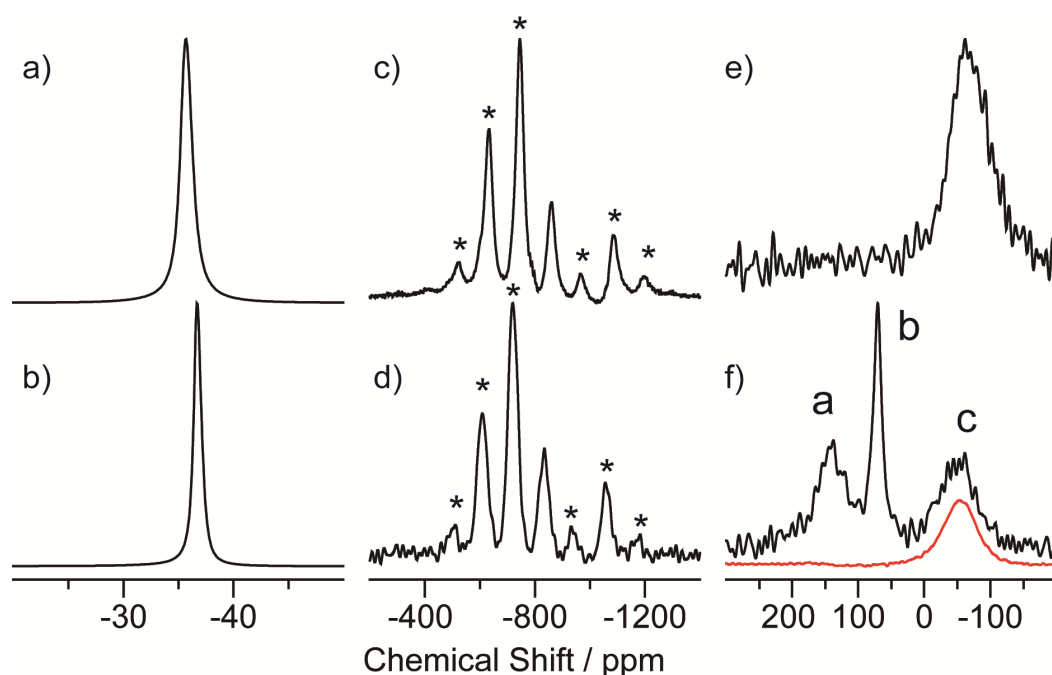


Figure 4. ^{19}F (a, b), ^{119}Sn (c, d), and ^{207}Pb (e, f) MAS NMR spectra of the β (a, c, and e) and β^+ (b, d, and f) samples at ca. 40 °C. An exponential line broadening of 1000 Hz was applied to ^{119}Sn (c, d) spectra prior to Fourier transformation. Except for the ^{207}Pb spectrum in red, which was obtained by the ^{19}F to ^{207}Pb CP method, all spectra were taken by using the single-pulse sequence without the Hahn echo for ^{19}F , and with the Hahn echo for ^{119}Sn and ^{207}Pb . The peaks marked by the asterisk are the spinning sidebands. The three signals in the ^{207}Pb spectrum of the β^+ sample (f) are labeled by "a" to "c" as indicated in the figure.

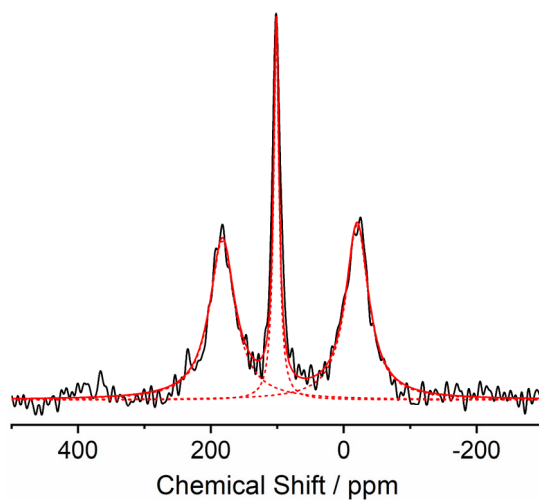


Figure 5. Spectral decomposition of the ^{207}Pb MAS spectrum of $\beta^+\text{-PbSnF}_4$ at 80°C (the black line) by three Lorentzian lineshapes (the red dotted lines). The sum of the three Lorentzian lineshapes is given by the red solid line. For quantitative analysis, the pulse-repetition time 13 s was set to be much longer than the spin-lattice relaxation times of the three signals (Pb-a: ca. 1.2 s, Pb-b: ca. 1.6s, and Pb-c: ca. 1.9 s).

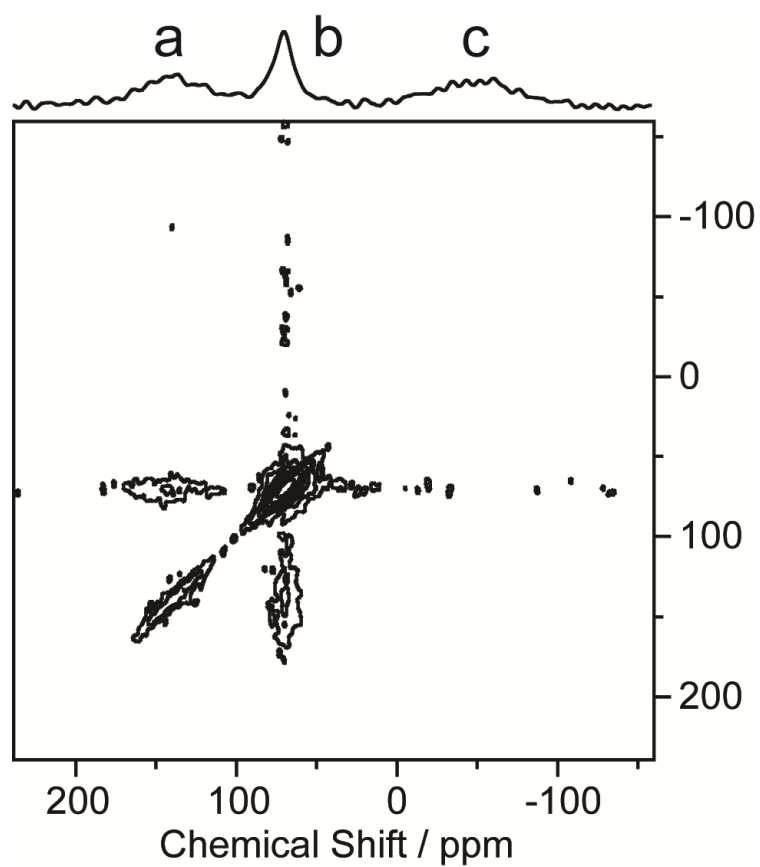


Figure 6. ^{207}Pb - ^{207}Pb dipolar-correlation RFDR spectrum of β^+ - PbSnF_4 ($x = 1.21$) with the mixing time of 9.6 ms at room temperature.

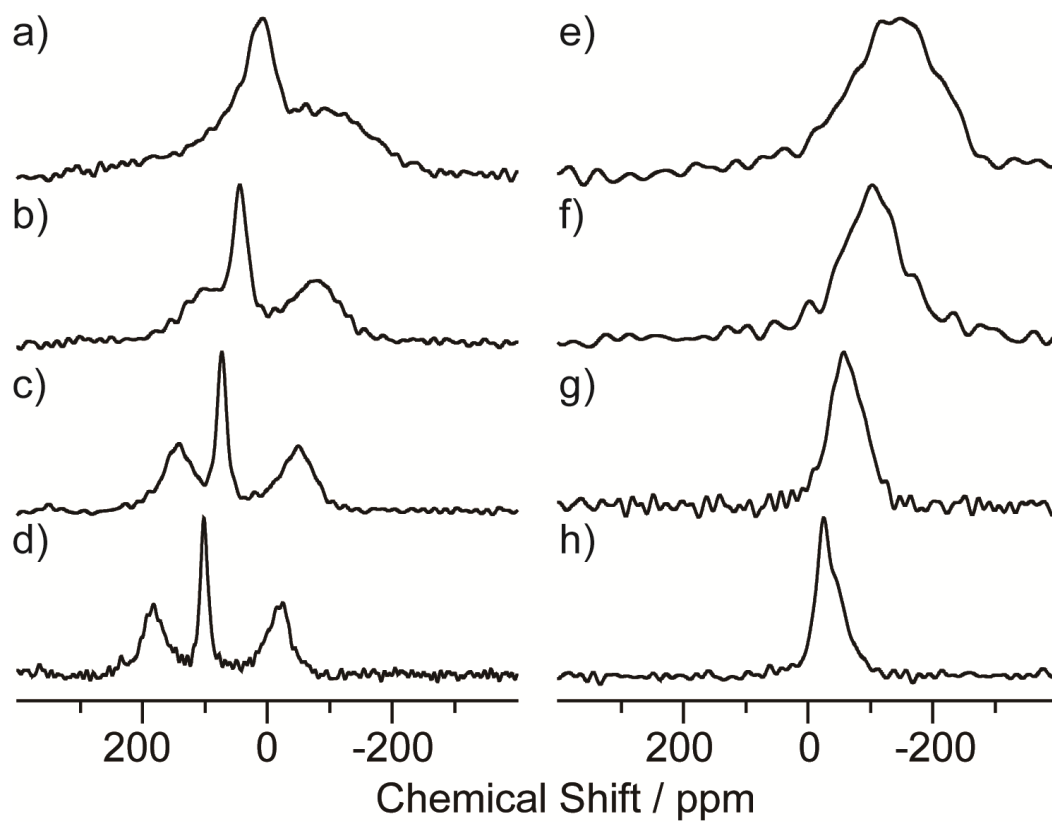


Figure 7. ^{207}Pb MAS-NMR spectra of the β^+ (a-d) and β (e-h) samples observed at $-40\text{ }^{\circ}\text{C}$ (a, e), $0\text{ }^{\circ}\text{C}$ (b, f), $40\text{ }^{\circ}\text{C}$ (c, g), and $80\text{ }^{\circ}\text{C}$ (d, h).

Supporting Information

High Anionic Conductive Form of $\text{Pb}_x\text{Sn}_{2-x}\text{F}_4$

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Contents

Figure S1. Observed (red cross) and calculated (blue line) neutron diffraction profile for β^+ - PbSnF_4 ($x = 1.15$).

Figure S2. Figure S2. Temperature dependence of the ^{207}Pb chemical shifts in β^+ - and β - PbSnF_4 obtained by spectral decomposition.

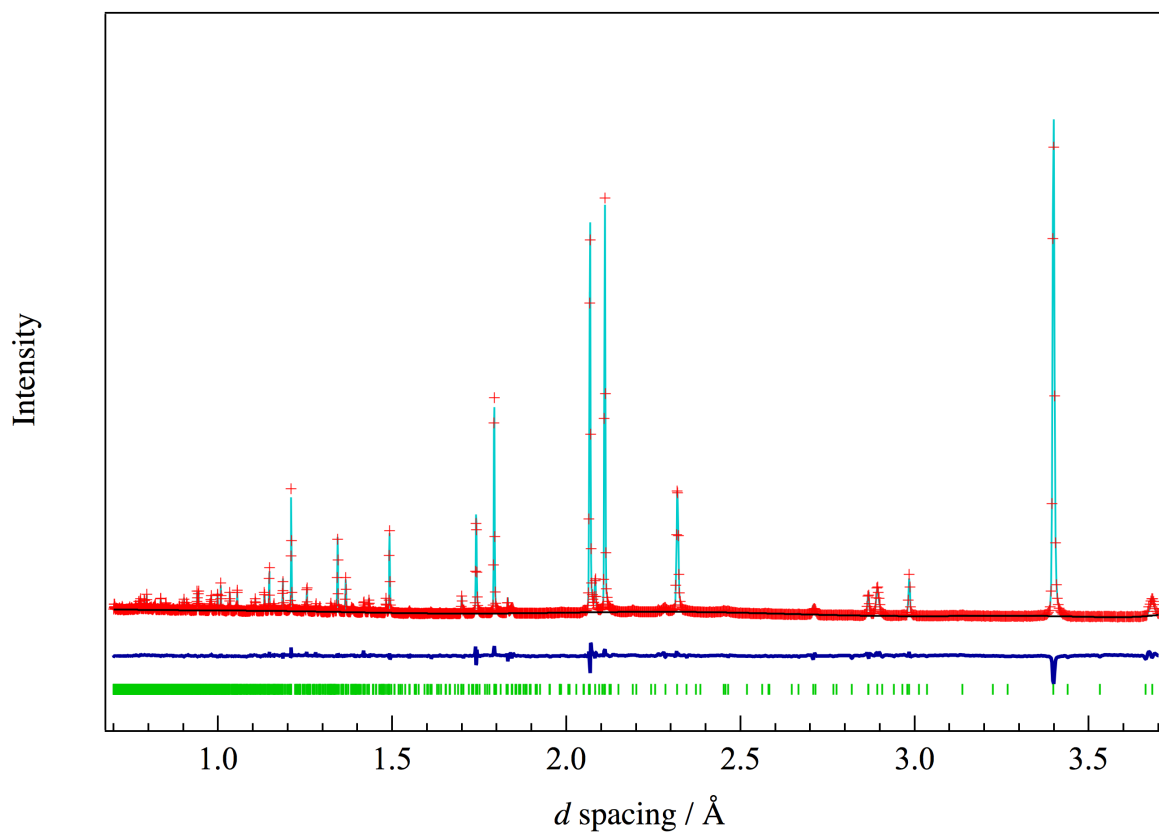


Figure S1. Observed (red cross) and calculated (blue line) neutron diffraction profile for β^+ -PbSnF₄ ($x = 1.15$). A difference curve (observed minus calculated, blue line) is shown at the bottom of figure; under the diffraction profiles, reflection positions are indicated by green tick marks. The cell parameters are as follows: space group $P4/nmm$, $a = 4.22030(3)$ Å, $c = 51.59771(76)$ Å, volume = 919.0023(16) Å³.

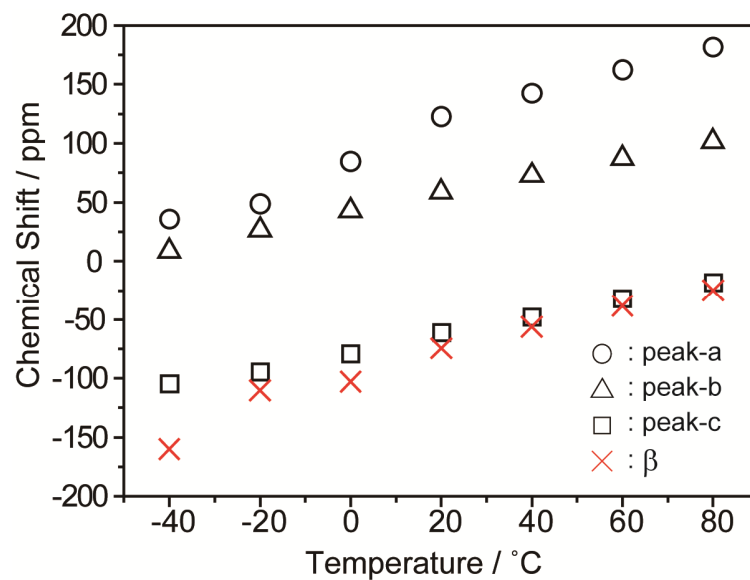


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